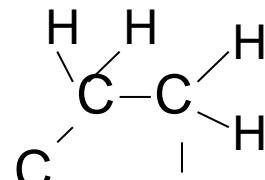
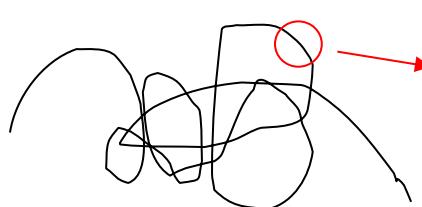


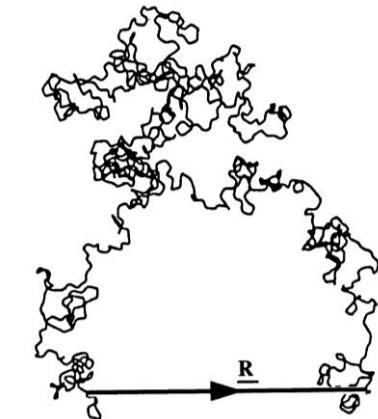
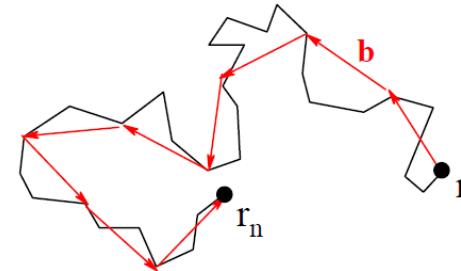
Basic concepts in Soft Matter (additional slides)

Molecular models of polymer chains: Ideal chain (non-interacting, ‘fantom’ solvent)

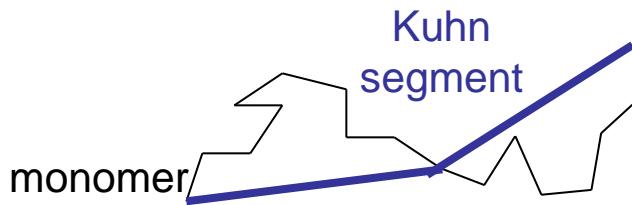
Definition of the chain configuration:



Example: polyethylene ($\text{CH}_2)_n$



Approximation: the *Kuhn segment* (and equivalent chain)



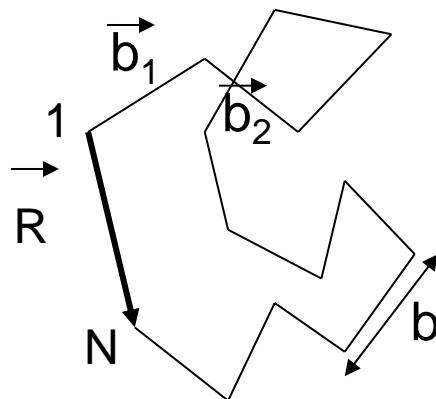
W. Kuhn 1899-1963

- No volume, no interaction between the segments
- equivalent chain with N Kuhn segments
- each with fixed length (= b)
- Bond between two monomers requires specific angle. Bond between two Kuhn segments can take any angle value (freely joint).

This approximation allows us to use the so-called Random walk model

Molecular models of polymer chains: average end-to-end distance

Random walk model:



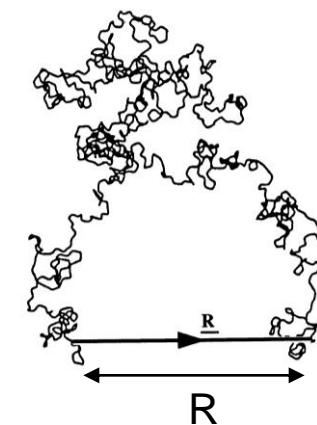
$$\left\{ \begin{array}{l} \vec{R} = \sum_{i=1}^N \vec{b}_i \\ \langle \vec{R} \rangle = \sum_{i=1}^N \langle \vec{b}_i \rangle = 0 \\ \langle \vec{R}^2 \rangle = \sum_{i=1}^N \sum_{j=1}^N \langle \vec{b}_i \vec{b}_j \rangle = Nb^2 \end{array} \right.$$

$$R^2 = C_\infty N b^2$$

$$\langle |R|^2 \rangle = R^2 = Nb^2$$

Quadratic distance R^2 :

$$\langle \vec{R}^2 \rangle = \langle R^2 \rangle = \frac{1}{p} \sum_{i=1}^p \vec{R}_i^2 \quad p: \text{all possible configurations}$$

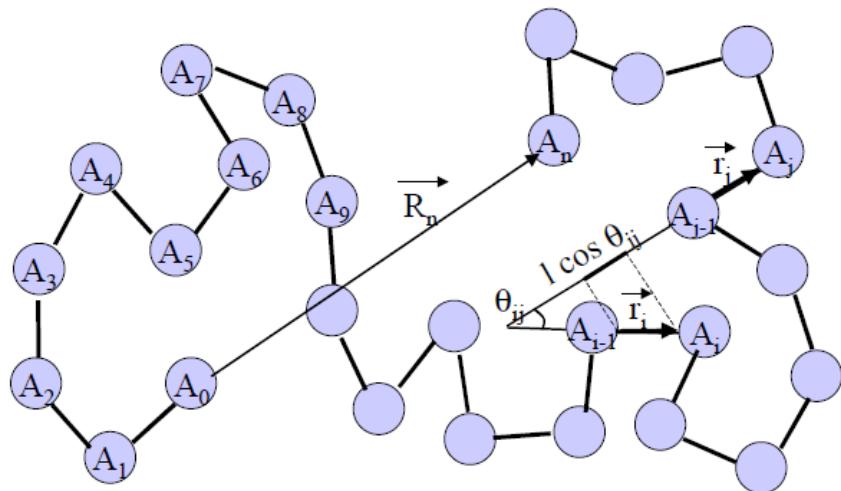


For each configuration:

$$\vec{R}_i = \sum_{j=1}^N \vec{b}_j \rightarrow R_i^2 = \sum_{j=1}^N \vec{b}_j \vec{b}_j + 2 \sum_{j < k} \vec{b}_j \vec{b}_k = \sum_{j=1}^N \vec{b}_j \vec{b}_j + 2 \sum_{j < k} b^2 \cos(\theta_{jk})$$

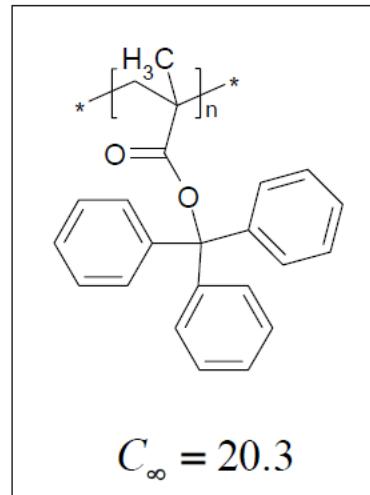
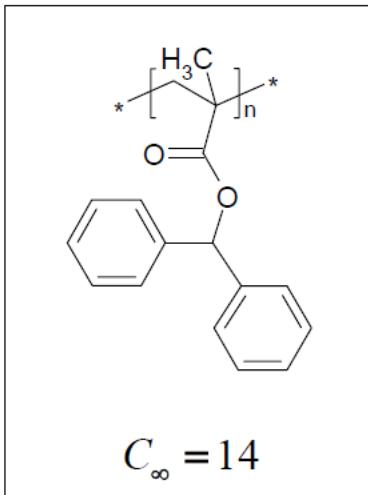
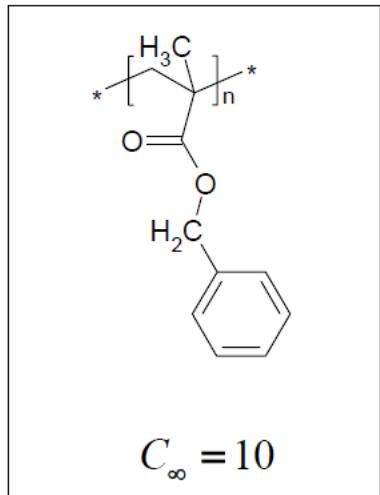
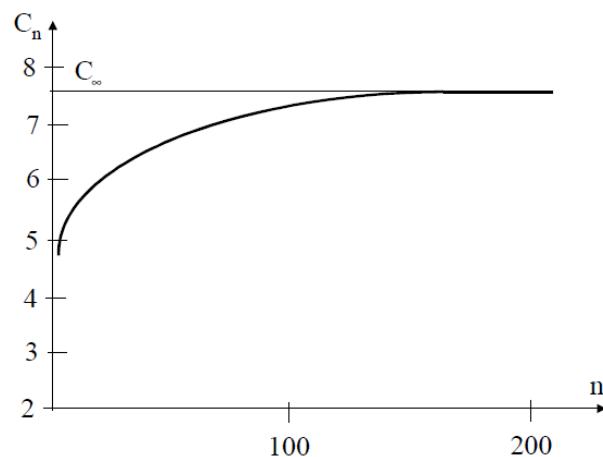
$$\vec{R}_i^2 = R_i^2 = \vec{R}_i \cdot \vec{R}_i = Nb^2 = 0$$

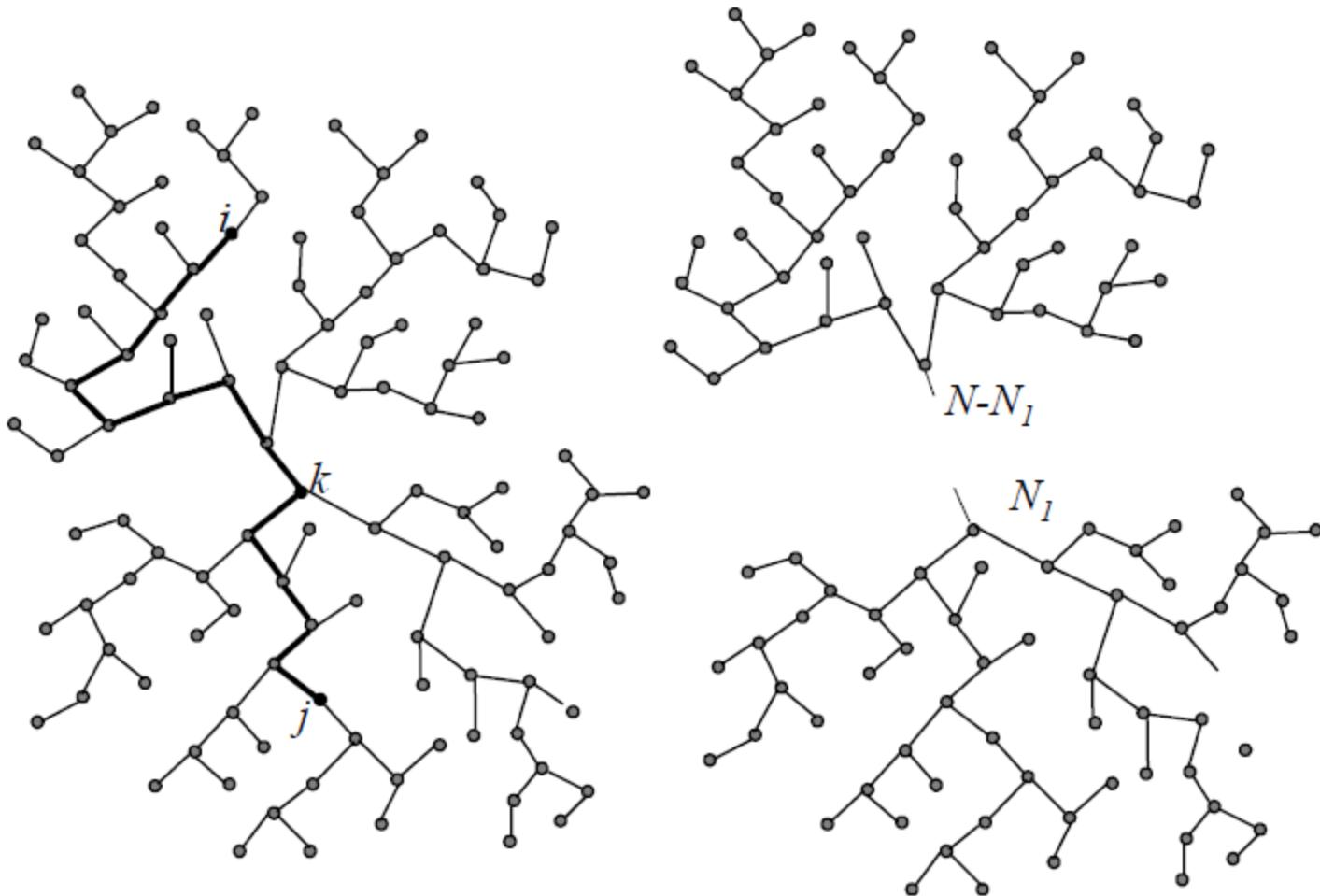
Chemical chain with N_m monomers, each of size m , or N Kuhn segments each of b

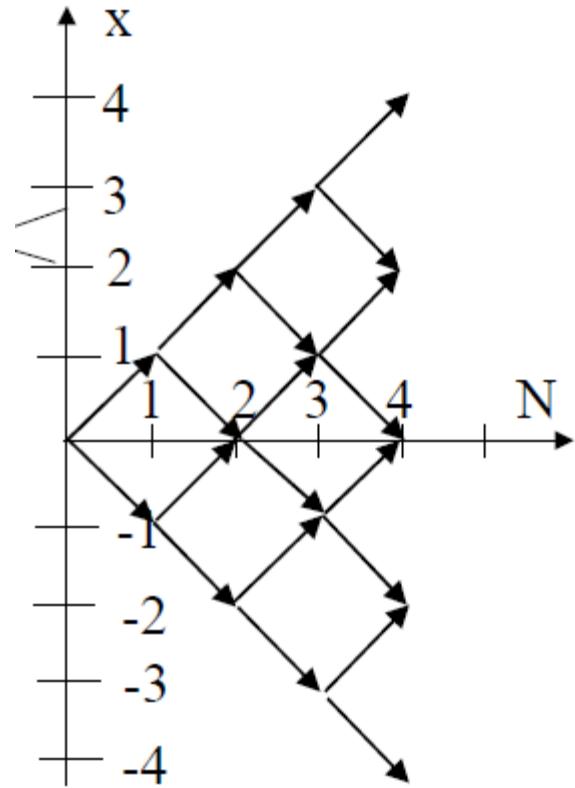


Flory's characteristic ratio

$$C_n = \frac{1}{n} \sum_{i=1}^n C'_i \quad C'_i \equiv \sum_{j=1}^n \langle \cos \theta_{ij} \rangle$$







$W(N, x)$

Different possible trajectories from start to position x with N steps

Example: $x=2$ reached with 0 ways if $N=1$, 3, with 1 if $N=2$, with 4 if $N=4$;
 $x=-1$ reached with 1 way if $N=1$, with 0 if $N=2$, 4 and with 3 if $N=3$

$$N = N_+ + N_-$$

$$x = N_+ - N_-$$

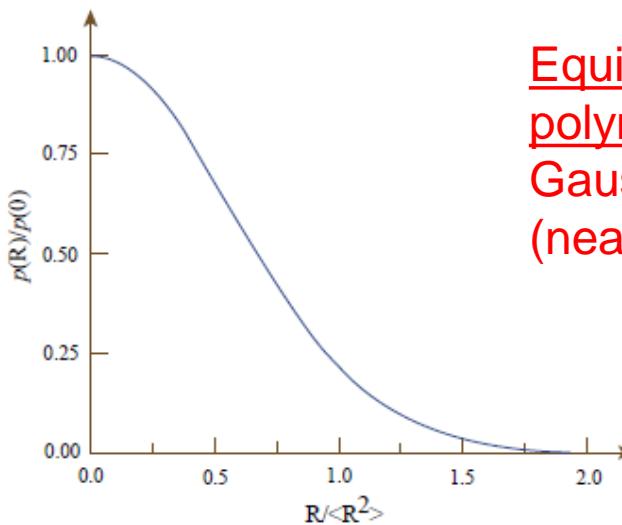
Single Gaussian chain (conformation): Size distribution

Random walk statistics (same step R)

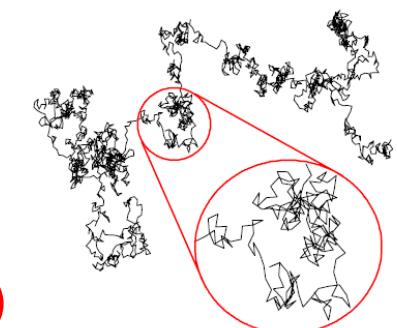
Probability density function (for end-end distance):

$$P(\underline{R}_0 = \underline{R}) \propto \exp\left(\frac{-3|\underline{R}|^2}{2Nb^2}\right)$$

(Gaussian)



Equilibrium form of polymer chain:
Gaussian coil
(nearly spherical)



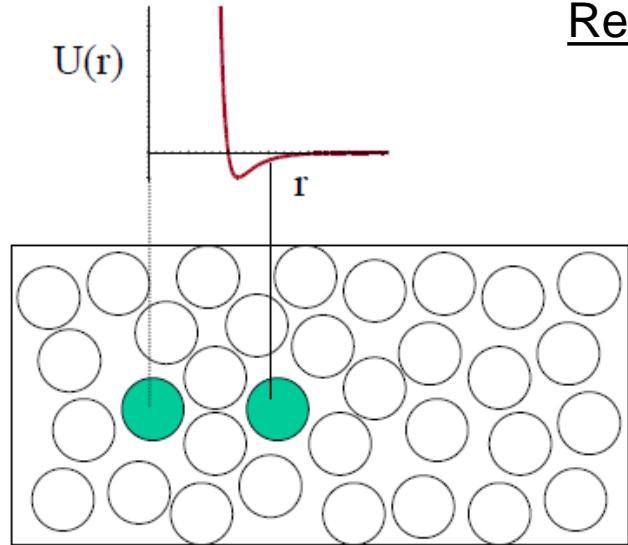
Compare:

$$V = Nb^3 \text{ vs. } V = R^3 = N^{3/2}b^3$$

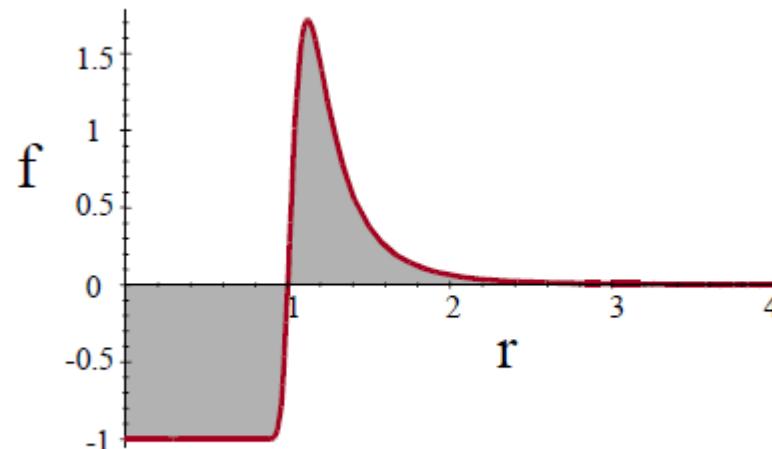
Polymers are fractal objects (here for ideal chain, $d_f=2$)

The effects of solvent quality (temperature)

Monomer pair interaction potential in solution; Boltzmann factor; f-Meyer function; excluded volume



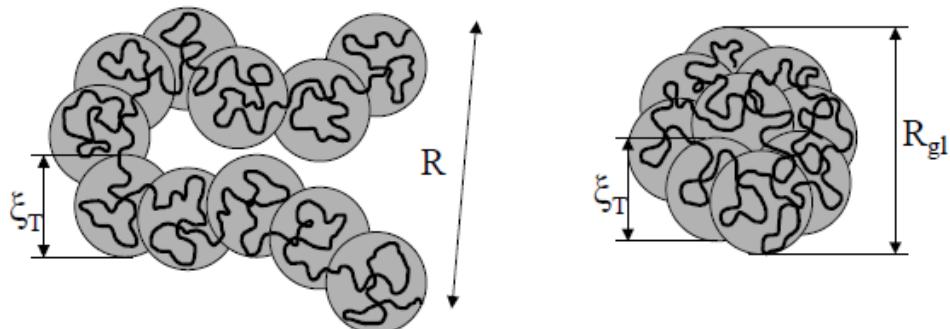
Real chains



$$v = - \int f(\vec{r}) d^3r = \int (1 - \exp[-U(r)/(kT)]) d^3r$$

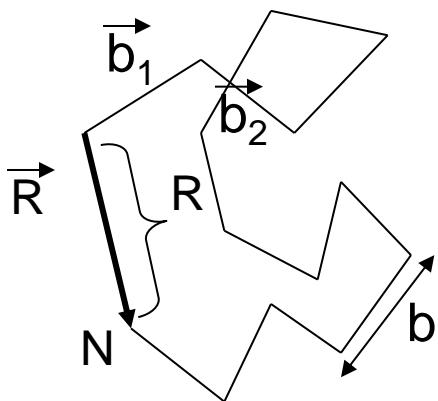
Athermal, good: $v>0$; theta: $v=0$; bad, non-solvent: $v<0$

Athermal: $v_{\max} = b^3$ non-solvent: $v_{\min} = -b^3$



Thermal blob $\xi_T \approx bg_T^{1/2}$

The effects of solvent quality (temperature)



Theta solvent (T_θ): $R \sim N^{1/2}$

P. J. Flory
1910-1985

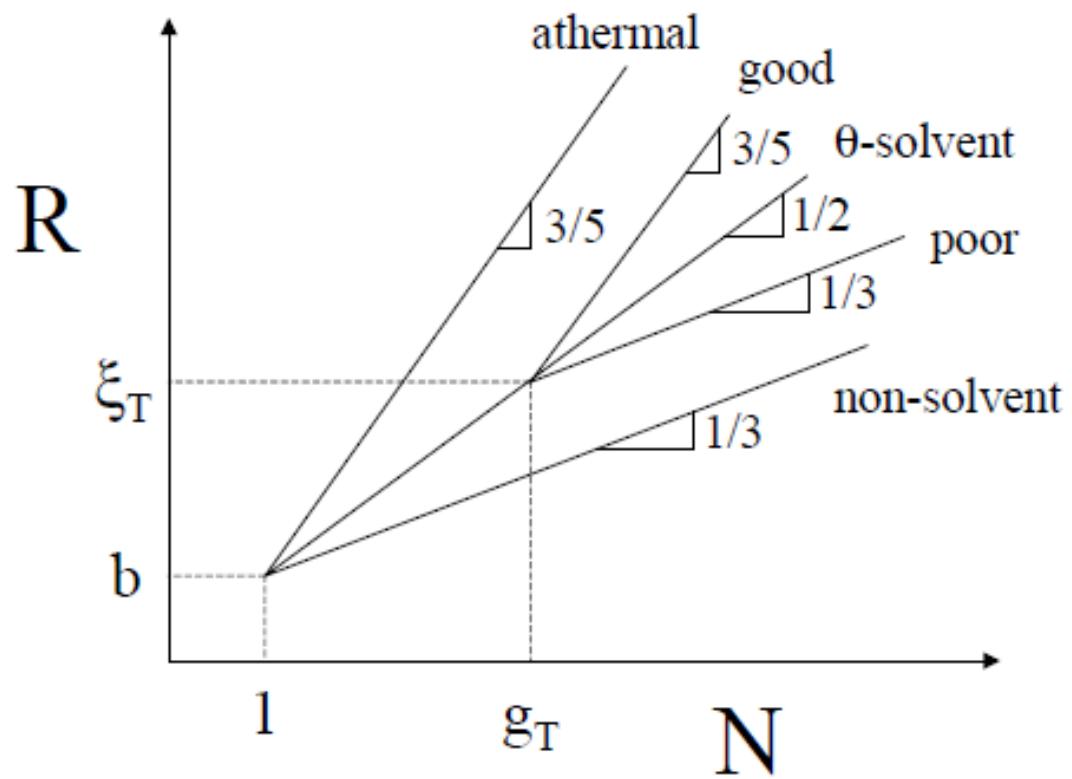
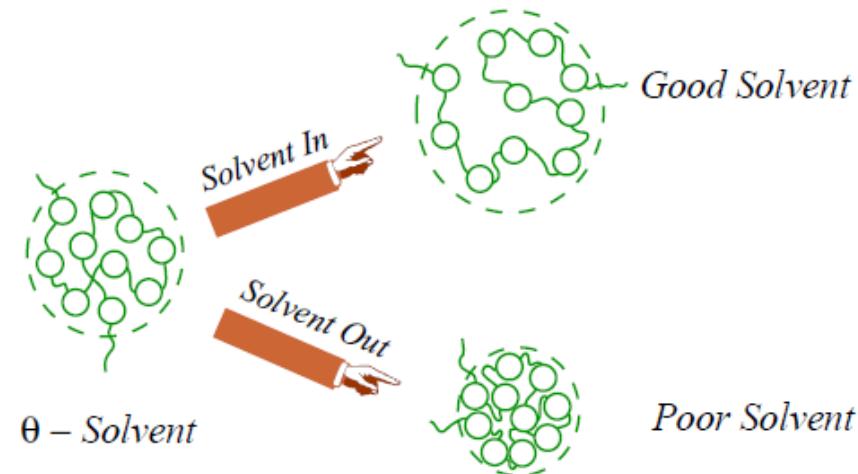
Interactions:

- { Good solvent: $T > T_\theta$, $R \sim N^{3/5}$ swelling
- Poor solvent: $T < T_\theta$, shrinkage (phase separation)
- Range: Athermal, good, theta, bad, non-solvent

Key idea: blobs, excluded volume

Approach: minimize free energy to get size

$$F = F_{int} + F_{ent} \approx kT \left(v \frac{N^2}{R^3} + \frac{R^2}{Nb^2} \right) \quad \frac{\partial \Delta F}{\partial R} = 0 \quad R_F \approx v^{1/5} b^{2/5} N^{3/5}$$



Application:

thermoreactive polymers (e.g., PNIPAM microgels)

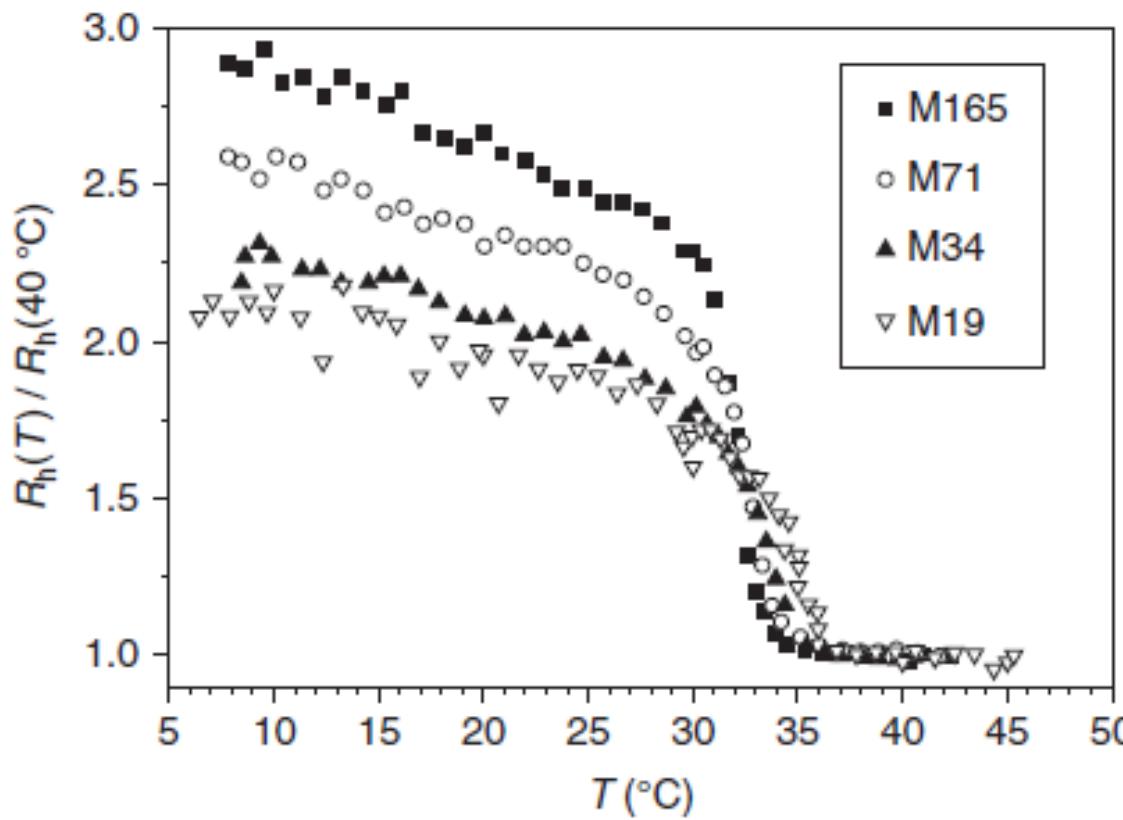
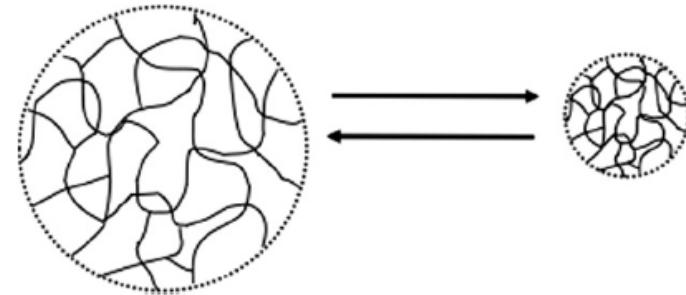
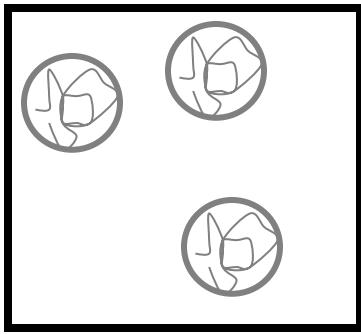
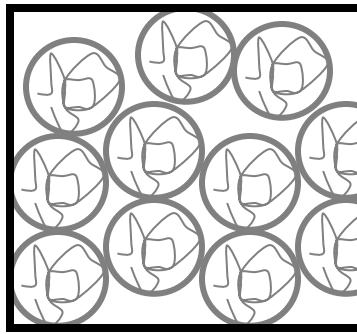


Figure 18 Size of PNIPAM microgels with different cross-linker content (increasing from top to bottom) vs. temperature. Reproduced with permission from Figure 1 in Senff, H.; Richtering, W. *Colloid Polym. Sci. 2000, 278, 830.*¹⁴¹

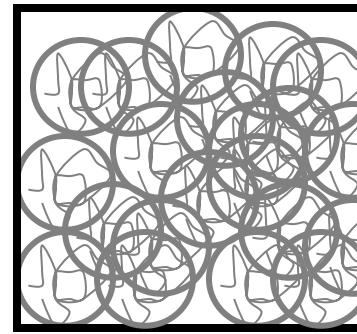
III. Polymer: From diluted to concentrated state



$c < c^*$



$c = c^*$



$c > c^*$

$$c^* = \frac{M}{N_a \frac{4}{3} \pi R^3}$$

M: [g/mol]

c^* : mass per volume

Dilute solutions:

Concentrated solutions or melts:

$$\phi_{eff} = \frac{c_{Mass}}{c^*} = c \frac{4}{3} \pi R^3 \ll 1$$

$$c = \frac{c_{Mass} N_a}{M}$$

c_{Mass} : mass per volume

c: number of molecules per volume

$c_{Mass} > c^*$: Interpenetration of the chains

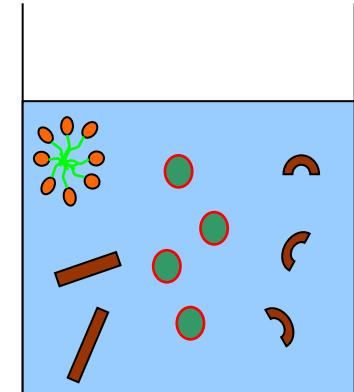
c^* : overlap concentration

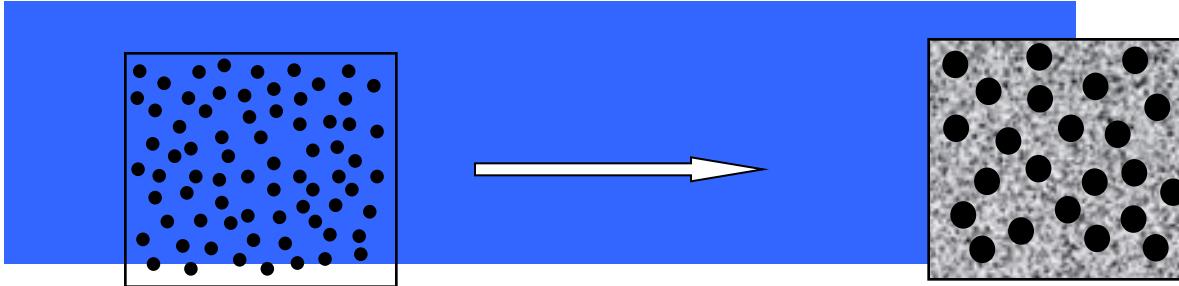
Semi-dilute solutions:

$c_{Mass} \approx c^*$

Size of dispersed particles: $\sim 10 \text{ nm}$ to $\sim 5 \mu\text{m}$

Brownian motion keeps $\Rightarrow k_B T > m_B g R$
them from sinking $\Rightarrow \text{radius } R \leq 1-5 \mu\text{m}$





Atomic

X1000

Colloidal

Time scales

$$\langle \Delta r(t)^2 \rangle = 6Dt$$

Slow

$$t \approx 1\text{ms} \dots 1\text{s}$$

Stokes-Einstein-Sutherland diffusion

$$(\text{Atomic : } t \approx 10^{-12} - 10^{-10} \text{s})$$

Mechanical response

$$G \propto \frac{k_B T}{R^3}$$

Soft

$$G \approx 1 - 1000 \text{ Nm}^{-2}$$

$$(\text{Metals: } G \approx 10^9 - 10^{12} \text{ Nm}^{-2})$$

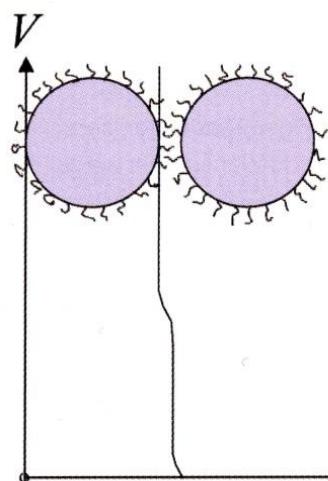
Length scale

$$d \approx \lambda$$

(wavelength of light)

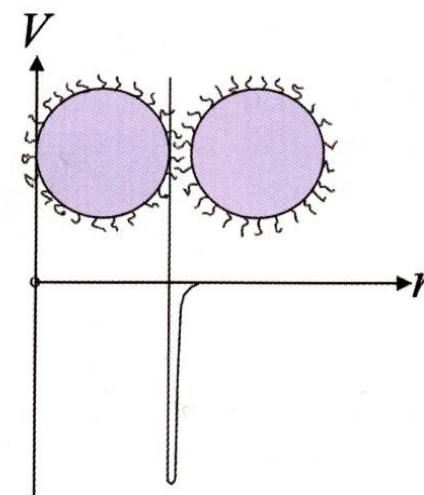
Large Enough =>
good for microscopy
and light scattering

Forces - Interactions



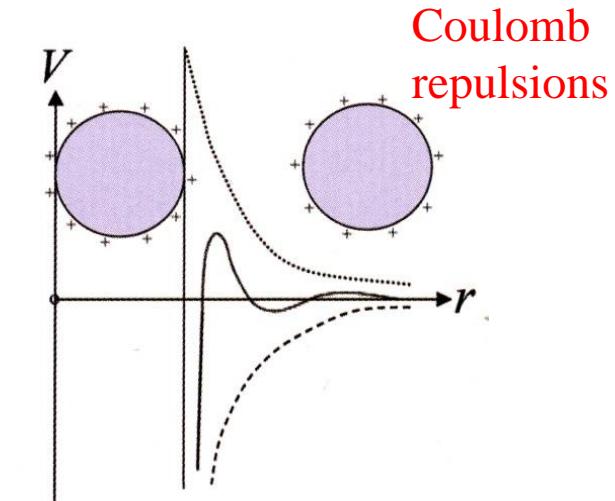
a)

nearly hard spheres



b)

Attractive (sticky spheres)

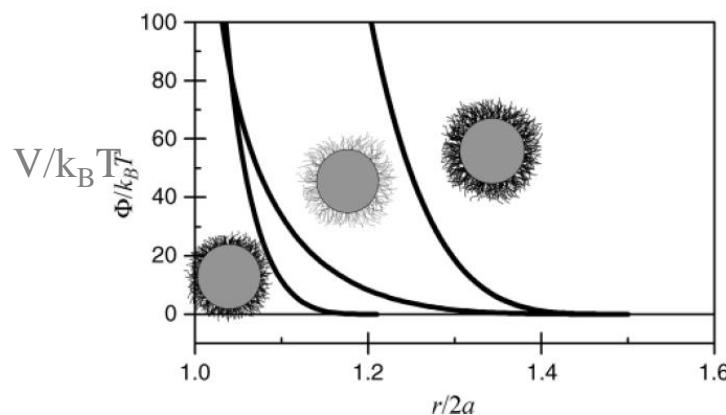


c)

Coulomb
repulsions

Van der Waals
attractions

In bad solvent



Sterically stabilized

Charged colloids
(DLVO potential)

(G. Petekidis)

Scale of main forces

- **Gravitational:** $F_{gravity} \approx R^3 \Delta \rho g$

- **Brownian:** $F_{Brownian} \approx k_B T / R$

- **Electrostatic:** $F_{coulomb} \approx \epsilon \epsilon_0 \zeta^2$

- **Viscous (Stokes drag):** $F_{viscous} \approx \eta R v$

- **Van der Waals:** $F_{VdW} \approx A_{eff} / R^2$

- **Inertia:** $F_{inertia} \approx \rho R^2 v^2$

Example :

$$R = 1 \mu\text{m}, \eta = 1 \text{cp} = 10^{-3} \text{ Pa s}$$

$$\rho = 10^3 \text{ kg/m}^3, \Delta \rho / \rho = 0.01$$

$$T = 20^\circ\text{C}, v = 1 \mu\text{m/s}$$

$$A_{eff} = 10^{-20} \text{ Joule}, \zeta = 50 \text{ mV}$$

$$g = 10 \text{ m/s}^2, \epsilon = 100, \epsilon_0 = 8.85 \cdot 10^{-12} \text{ C/Vm}$$

rations of forces:

$$\frac{F_{coulomb}}{F_{Brownian}} \approx 100$$

$$\frac{F_{vdW}}{F_{Brownian}} \approx 1$$

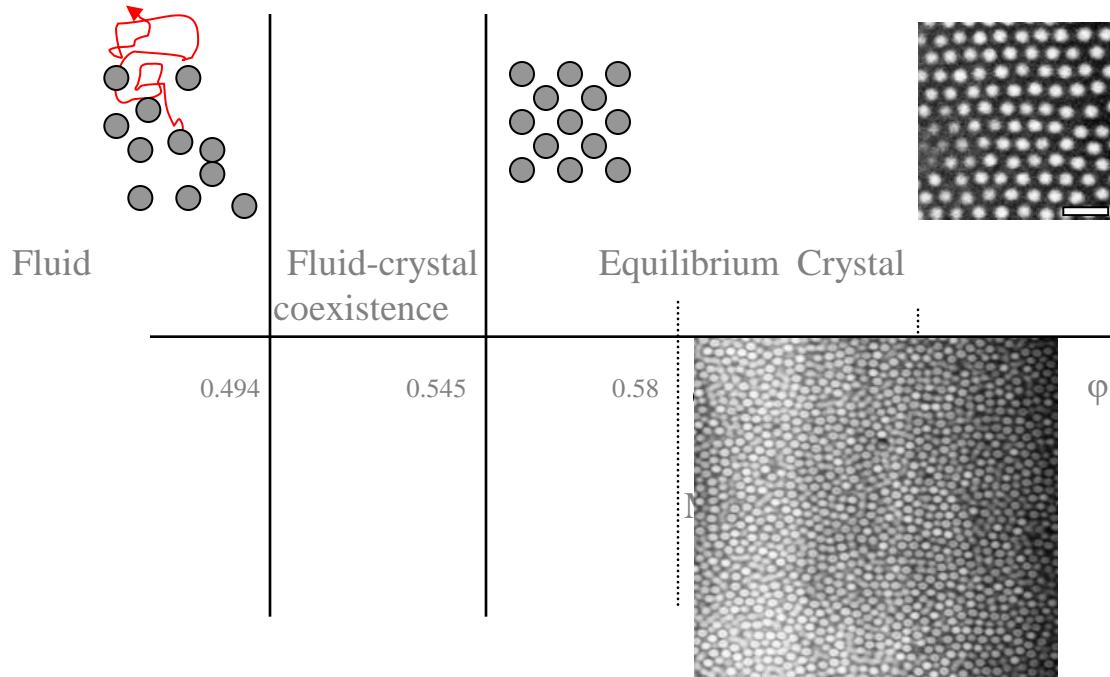
$$\frac{F_{viscous}}{F_{Brownian}} \approx 1$$

$$\frac{F_{gravity}}{F_{viscous}} \approx 0.1$$

$$\frac{F_{inertia}}{F_{viscous}} \approx 10^{-6}, (=Re)$$

Phase behavior – Brownian Hard Spheres

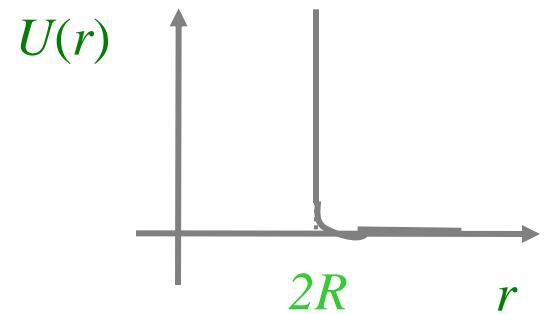
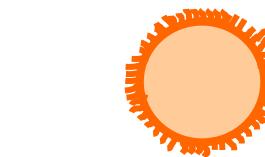
One-dimensional phase diagram



Computer simulations:
Alder & Wainwright(1957)
Wood & Jacobson (1957)
Hoover & Ree (1968)

$$\phi = \frac{N}{V} \cdot \frac{4}{3} \pi R^3$$

PMMA spheres
Sterically stabilized
with PHSA

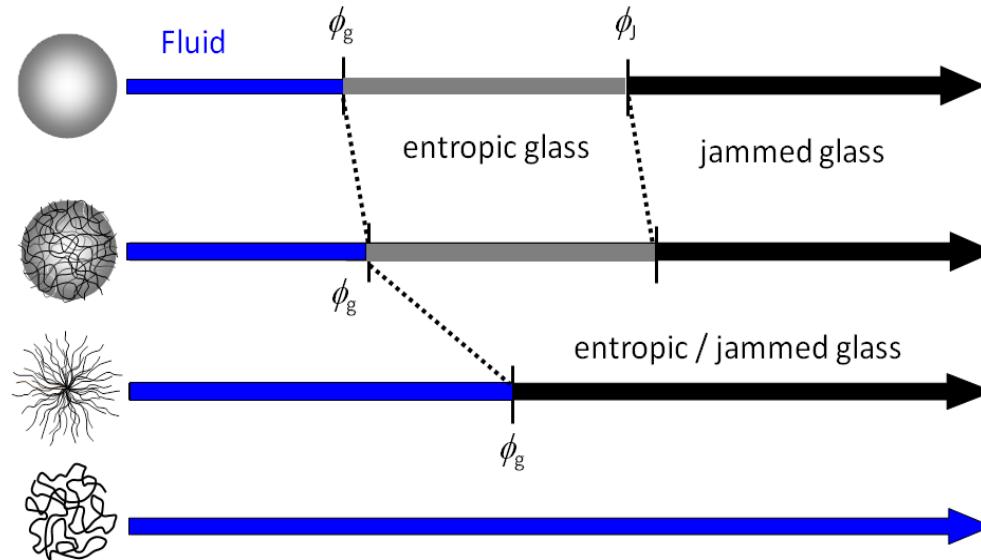


Entropy Driven Crystallisation
Crystal has *higher* entropy
than metastable fluid
at same concentration

Experiments: Pusey, van Megen, Nature, 1986

(G. Petekidis)

Softness: state transitions



Challenge: determine volume fraction $\Phi_{\text{eff}} = C/C^*_h = nV_0$

Poon et al. SM 2012

Conley et al. Sci. Adv. 2017

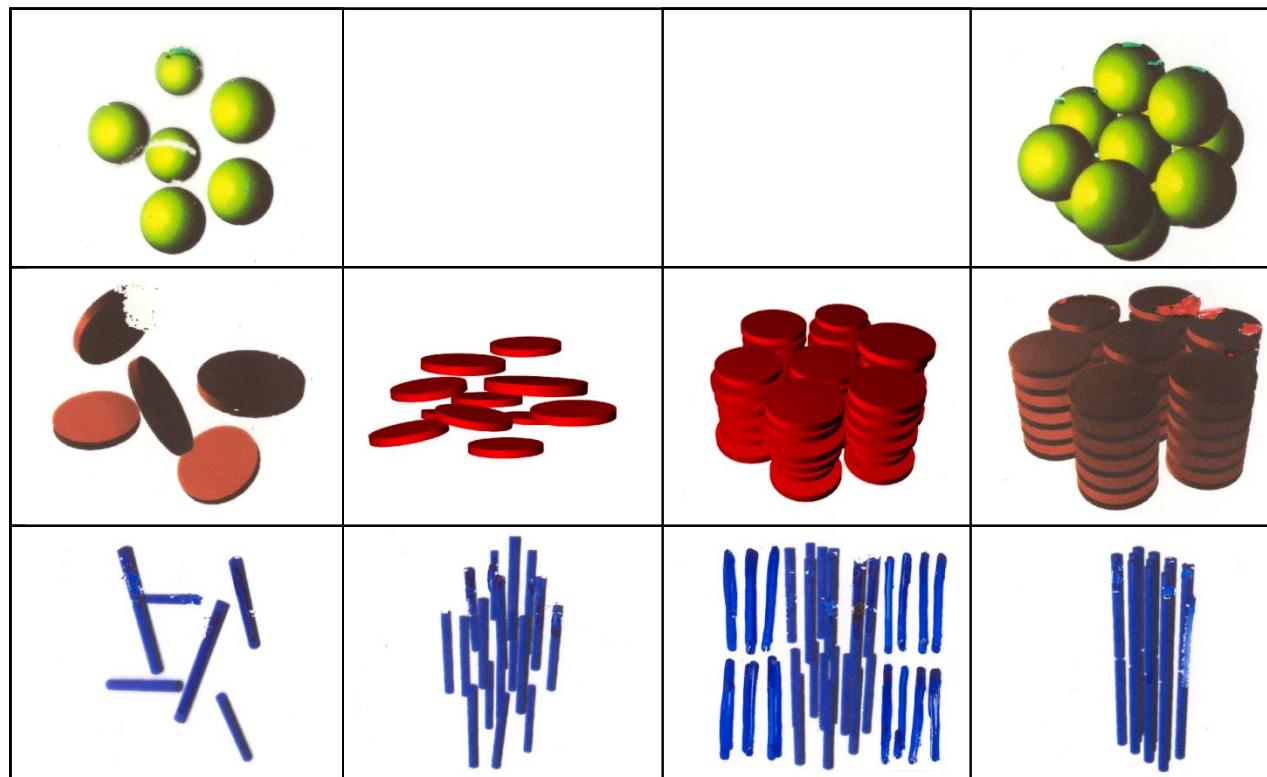
Bouhid de Auiar et al. Sci. Rep. 2017

van der Scheer et al. ACS Nano 2017

Shape matters

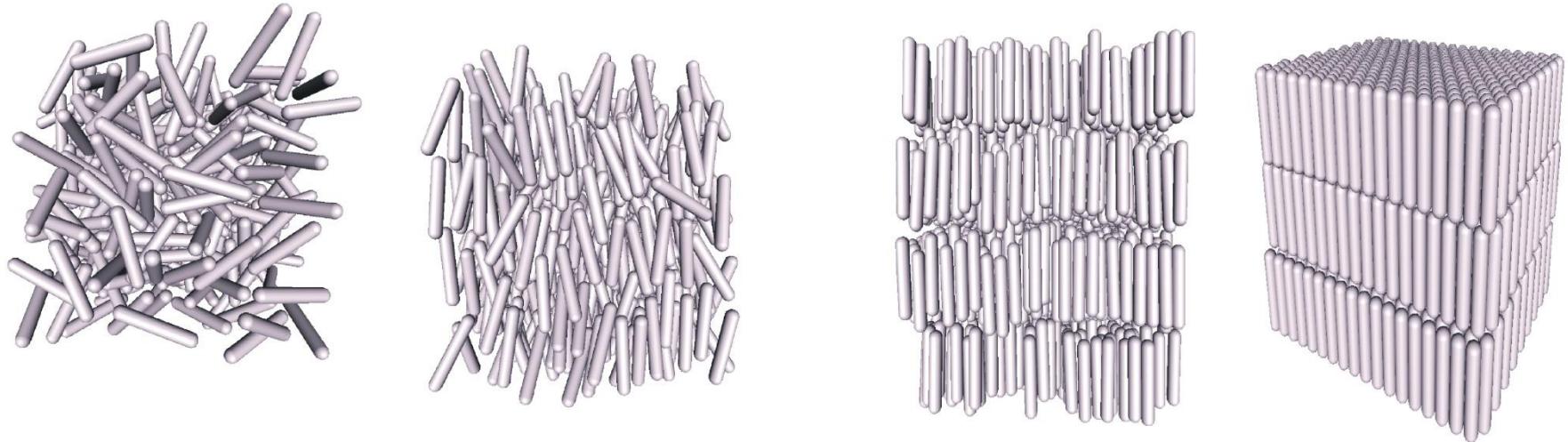
isotropic

crystal



concentration

Other entropy driven transitions: Liquid crystals



Isotropic liquid

Nematic-liquid crystals

Smectic-Liquid
Crystals

Crystal



Decreasing temperature (thermotropic)

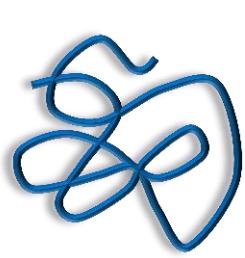
or increasing concentration (Lyotropic)

Entropy driven ordering: Isotropic-nematic transition in Hard rods

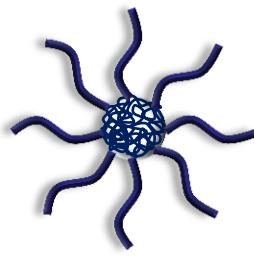
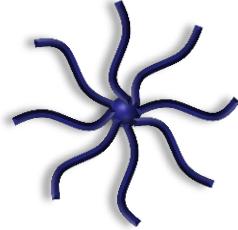
Softness: from polymers to colloids

$$\varepsilon = \frac{\Delta F}{k_B T}$$

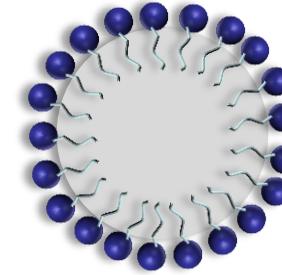
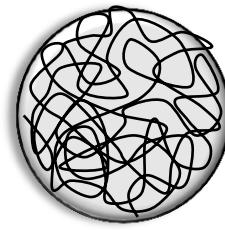
$$\varepsilon \approx 1$$



Polymer coil



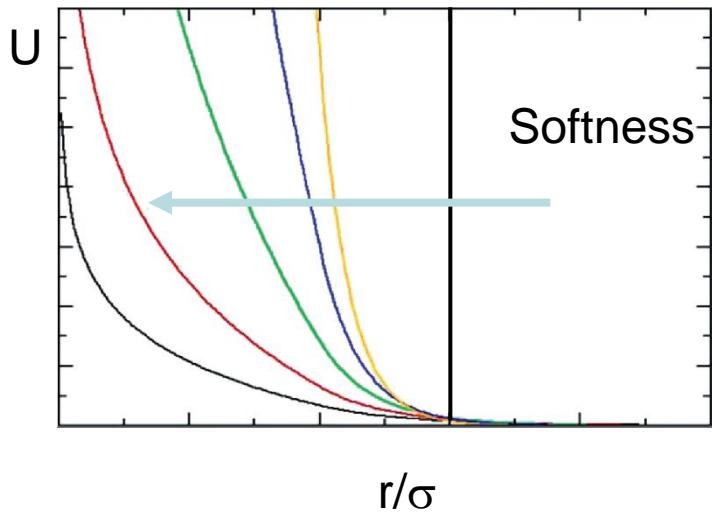
volume fraction > 1



$$\varepsilon = \frac{ER^3}{k_B T} \rightarrow \infty$$



Hard sphere



$$V_{Hertz}(r) = \begin{cases} \epsilon \left(1 - \frac{r}{R_i + R_j}\right)^{\frac{5}{2}} & \text{for } r < R_i + R_j \\ 0 & \text{for } r > R_i + R_j \end{cases}$$

$$\frac{V_{star}(r)}{k_B T} = \begin{cases} (5/18)f^{3/2} \left[-\ln(r/\sigma) + (1 + \sqrt{f}/2)^{-1} \right] & \text{for } r \leq \sigma \\ (5/18)f^{3/2}(1 + \sqrt{f}/2)^{-1} (\sigma/r) \exp[-\sqrt{f}(r - \sigma)/2\sigma] & \text{for } r > \sigma \end{cases}$$